

## Trans와 Cis-1,4-Cyclohexanedimethanol, Terephthaloyl Dichloride, 2,6-Naphthalenedicarboxylic Chloride로부터 합성된 폴리에스터 공중합체의 단일 단계 용액 중합과 열적 특성

Fiaz Hussain · 정재민 · 박상원 · 강수정 · 김진환<sup>†</sup>

성균관대학교 고분자공학과

(2019년 2월 26일 접수, 2019년 3월 15일 수정, 2019년 3월 15일 채택)

### Single-Step Solution Polymerization and Thermal Properties of Copolyesters Based on High Trans-1,4-Cyclohexanedimethanol, Terephthaloyl Dichloride, and 2,6-Naphthalene Dicarboxylic Chloride

Fiaz Hussain, Jaemin Jeong, Sangwon Park, Soo-Jung Kang, and Jinhwan Kim<sup>†</sup>

Department of Polymer Science and Engineering, Sungkyunkwan University,  
300 Cheoncheon-dong, Jangan-gu, Suwon, Gyeonggi 16419, Korea

(Received February 26, 2019; Revised March 15, 2019; Accepted March 15, 2019)

**초록:** Poly(1,4-cyclohexylenedimethylene terephthalate-co-1,4-cyclohexylenedimethylene 2,6-naphthalenedicarboxylate) (PC#TN#) 폴리에스테르 공중합체를 안정제 및 금속 촉매를 넣지 않고 실온에서 단일 단계 용액 중합법으로 합성하였다. 얻어진 PC#TN# 폴리에스테르 공중합체의 화학 구조는 <sup>1</sup>H NMR로 규명하였고, 열특성은 TGA와 DSC로 확인하였다. 열적 특성은 diacid 부분의 나프탈렌 단위와 diol 부분의 cis/trans-1,4-cyclohexanedimethanol(CHDM) 이성질체의 비율을 변화시킴으로써 조절하였다. trans-CHDM의 증가된 함량은 유리 전이 온도( $T_g$ ) 및 용융 온도( $T_m$ )를 증가시킬 수 있었다. 나프탈렌 단위의 함량을 증가시킴으로써  $T_g$ 는 선형적으로 증가하는 반면,  $T_m$ 은 감소하다가 공융점(2,6-naphthalene dicarboxylic acid 36 mole%)에 도달하면 증가하였다. 합성된 폴리에스테르 공중합체의 열적 분해 거동은 나프탈렌 및 trans-CHDM 단위를 증가시킴으로써 상당히 개선되었다.

**Abstract:** In this work, two series of poly(1,4-cyclohexylenedimethylene terephthalate-co-1,4-cyclohexylenedimethylene 2,6-naphthalenedicarboxylate) (PC#TN#) copolyesters were synthesized by a simple one-step solution polymerization method at room temperature without the use of stabilizer and metallic catalyst. Chemical compositions of obtained PC#TN# copolyesters were confirmed by nuclear magnetic resonance (<sup>1</sup>H NMR) spectroscopy. Thermal properties of synthesized copolyesters were determined by differential scanning calorimetry (DSC) and thermogravimetric analysis (TGA). Thermal properties of synthesized PC#TN# were controlled by varying the ratio of naphthalene units in diacid part and cis/trans-1,4-cyclohexanedimethanol (CHDM) isomers in diol part. An increased content of trans-CHDM can increase glass transition temperature ( $T_g$ ) and melting temperature ( $T_m$ ) profoundly. By increasing the content of naphthalene units,  $T_g$  increases linearly while  $T_m$  is first decreased then it starts to increase once the eutectic point (2,6-naphthalene dicarboxylic acid 36 mole%) is reached. Thermal degradation behavior of synthesized copolyesters was also significantly improved by increasing naphthalene and trans-CHDM units.

**Keywords:** 1,4-cyclohexanedimethanol, terephthalic acid, naphthalene units, solution polymerization, thermal properties.

## Introduction

Thermoplastic polyesters have gained the attention of academia and industrial researchers because of their wide range of

technical applications.<sup>1-6</sup> Synthesis of high molecular weight aliphatic polyesters was first reported by Carother and Hill.<sup>7</sup> But, inherent poor hydrolytic stability, low glass transition temperature ( $T_g$ ), and melting temperature ( $T_m$ ) of aliphatic polyesters strictly abandoned their commercial applications. Whinfield and Dickson reported the synthesis of poly(ethylene terephthalate) (PET) exhibiting acceptable thermal properties in 1949.<sup>8</sup> However, high crystallinity and low  $T_g$  of PET (80 °C)

<sup>†</sup>To whom correspondence should be addressed.  
jhkim@skku.edu, ORCID<sup>®</sup> 0000-0002-8126-4221  
©2019 The Polymer Society of Korea. All rights reserved.

limits its commercial applications at elevated temperatures.

Performance properties of PET can be improved by controlling the chemical structure of its backbone chain. Kibler *et al.* disclosed the synthesis process and thermal properties of poly(cyclohexane 1,4-dimethylene terephthalate) (PCT) in 1964.<sup>9</sup> When ethylene glycol of PET is replaced by 1,4-cyclohexanedimethanol (CHDM), the resulting PCT polyester have much higher  $T_g$  (88 vs 80 °C) and  $T_m$  (300 vs 260 °C).<sup>9</sup> For general plastic applications, processability of PCT polymer must be improved by modifying it with diacid or diol components. CHDM-modified PET copolyester and glycol-modified PCT copolyesters are amorphous in nature and they have better heat distortion temperature, higher impact strength, and superior chemical resistance than PET. These copolyesters have variety of commercial applications as injection molded polymers for medical and electronics applications.<sup>4,10</sup> In short, copolymerization depending upon different kinds of monomers and their content ratio directly influence thermal and crystallization behavior of synthesized copolyesters.<sup>10,11</sup> However, both high crystallinity and high melting temperature of PCT homopolymer act as obstacles for its commercial applications as a film.

Performance properties can also be improved by incorporating the thermally stable and rigid naphthalene units as diacid moiety into the PET backbone itself. When terephthalic acid (TPA) is replaced by the naphthalene dicarboxylic acid (NDA), the resulting polyester, poly(ethylene naphthalate) (PEN) have superior  $T_g$  (about 120 vs 80 °C) and  $T_m$  (270 vs 260 °C) than conventional PET. Soon after the discovery of PEN in 1969,<sup>12</sup> this polymer rapidly finds its applications as a performance materials in versatile areas.<sup>13</sup> PEN and its copolymers are considered to be strong candidates for high temperature applications.<sup>4,13,14</sup> However, high cost of monomer (NDA) used for the synthesis of PEN, birefringence, and necking behavior of PEN film during stretching process limit the wide range applications of this polymer. Crystallinity and thermal properties of PCT can also be controlled by introducing second diacid component into the molecular structure, these copolyesters are called acid-modified copolyesters (PCTA). Isophthalic acid (IPA) and cyclohexanedicarboxylic acid (CHDA) are the most common diacid modifiers. Introduction of small amount of second diacid widens the processing window of PCT polyester by lowering its  $T_m$ . However, these copolyesters are not suitable for the applications where good thermal properties are required and the higher amount of second diacid results in high cost and processing issues.<sup>10,15,16</sup>

Not only diol and diacid moieties but their stereochemistry

can also tune the thermal and mechanical properties of the copolyesters. Stereochemistry of CHDM (cis/trans isomers) can affect the comprehensive performance properties of CHDM based polymers. It is well established in the literatures that polymers based on trans-CHDM have superior barrier, thermal, mechanical, and chemical resistance when compared with their analogous polymers based on cis-CHDM.<sup>17-21</sup> For the first time in 1964, Kibler *et al.* discovered that PCT based on trans-CHDM have superior thermal properties than analogous polyester containing cis-CHDM isomers (310 vs 251 °C).<sup>9</sup> Berti *et al.* studied the effect of cis- and trans- isomers of diacid moieties and disclosed that performance properties, especially the thermal and barrier properties of copolyesters can be effectively improved by using monomers that contain high trans-isomers.<sup>22</sup> Young and Won tuned the thermal properties and crystal structure of poly(cyclohexane 1,4-dimethylene 2,6-naphthalate) (PCN) polymer by controlling the cis- and trans-configuration of CHDM.<sup>23</sup> It is suggested that molecular arrangements of copolyesters containing trans-CHDM isomers are more regular, stable, and symmetric than copolyesters based on high cis-CHDM isomers.<sup>18,21,23-25</sup>

Currently, almost all high temperature polyesters are synthesized by a complex two-step melt polymerization. First step involves the formation of pre-polymer by esterification reaction and the polymer of high molecular weight is obtained in the second step called polycondensation reaction. For the synthesis, prepolymer reacts with diol at relatively high temperature and pressure. By product removal set up is also attached at both steps of polymerization.<sup>11,15,26</sup> Due to harsh condition applied during melt polymerization a stoichiometric imbalance is observed due to degradation and sublimation of monomers and this phenomenon leads to the increment in side reactions and the reaction efficacy is reduced.<sup>27,28</sup> Some of the problems associated with melt polymerization are addressed by using suitable metallic catalysts.<sup>29,30</sup> However, titanium based catalysts, which are considered to be most effective catalysts among all metallic catalysts, induce yellow color into the synthesized product while antimony based catalysts are associated with some toxicity issues.<sup>31,32</sup> Generally, an additional thermal stabilizer is also used with metallic catalyst to prevent polymer degradation during the polycondensation and subsequent process which results in increased cost.<sup>33</sup> It is worthy to note that neither polymer degradation nor discoloration of the product is observed during solution polymerization reaction.<sup>34,35</sup> The major concern about the solution polymerization is the selection of pure solvent which facilitates the solubility of monomers and

easy recovery of synthesized polymer product. It is well established that pure and well defined polymers are synthesized by a reproducible one-step solution polycondensation process.

It would be a great effort to introduce a synthesis of novel thermally stable copolyesters by using the advantageous features of trans-isomers of diol (CHDM) and second dicarboxylic acid (NDA), simultaneously. To the best of our knowledge, only one industrial patent<sup>36</sup> is reported in which thermal and mechanical properties of copolyester film are controlled by utilizing the effect of stereochemistry of CHDM and second dicarboxylic acid (NDA). However, very limited information was provided in this patent. For the first time, we are reporting a detailed academia research showing the effect of both NDA (second diacid) and stereochemistry of diol (cis-/trans-CHDM) on the thermal properties of synthesized two series of copolymers, PC70TN# and PC100TN# containing 70 and 100% trans-CHDM isomers, respectively. They were synthesized by a short, simple, controlled, and straightforward one-step solution polymerization method. It is important to note that any kind of metallic catalyst or additional stabilizer is not used during the whole synthesis, so eliminating any kind of toxicity related issues. This study mainly focusses three parts: (1) introduction of a novel solution polymerization approach for the synthesis of two series of copolymers, at room temperature in the absence of toxic metallic catalysts and stabilizer; (2) characterization of synthesized copolyesters and (3) study the effect of stereochemistry of CHDM and naphthalene units on the thermal properties of synthesized copolyesters. So, in this study, the crystal structure and thermal properties of PCT homopolymers were controlled by the ratio of cis-/trans-CHDM isomers and the second dicarboxylic acid.

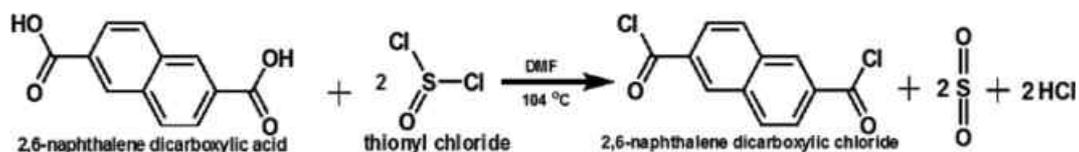
## Experimental

**Materials.** Reagent grade terephthaloyl dichloride (TPC,  $C_8H_4Cl_2O_2$ ), trans-1,4-cyclohexanedimethanol (100 mole% trans-isomer) ( $C_8H_{16}O_2$ ), and 4-dimethylaminopyridine (DMAP,  $C_7H_{10}N_2$ , 99% purity) were obtained from TCI (Japan). Extra pure (99.5%) tetrahydrofuran (THF,  $C_4H_8O$ ), sodium hydrox-

ide (NaOH), benzene ( $C_6H_6$ ), heptane ( $C_7H_{14}$ ), methyl alcohol ( $CH_3OH$ ), and thionyl chloride ( $SOCl_2$ ) were purchased from Samchun Chemical Co. Ltd (Korea). Phenolphthalein was purchased from Sigma Aldrich (China). 1,4-cyclohexanedimethanol (1,4-CHDM, 99.8 mole% purity) with 70 mole% trans-isomers was supplied by SK Chemicals (Korea). 2,6-naphthalenedicarboxylic acid (NDA,  $C_{12}H_8O_4$ ) was supplied by BASF (Germany). Dimethylformamide (DMF,  $C_3H_7NO$ ) was purchased from Duksan Chemicals (Korea). 2-Chlorophenol (OCP) was purchased from Junsei Chemical Co. Ltd (Japan). All chemical reagents were stored in the desiccator before use. Extra pure deionized water (DI  $H_2O$ ) was used during the whole synthesis process.

**Preparation of 2,6-Naphthalenedicarboxylic Chloride (NDC).** In order to improve the reactivity of NDA during the solution polymerization of two series of copolymers, it was converted into NDC ( $C_{12}H_6Cl_2O_2$ ) by following the procedure reported in the literature.<sup>37</sup> Schematic diagram for the modification of NDA into NDC is shown in Scheme 1. Thionyl chloride (4 mole) and 1.7175 g DMF (0.07618 mole) which was used as a catalyst were added into the solution of NDA (1 mole) in 375 mL benzene. The mixture of all reactants was heated slowly under gentle reflux up to 104 °C and these conditions were maintained for 24 h. Solution became clear upon the completion of reaction. It was cooled down to room temperature and the product in the form of yellow crystals was appeared. Obtained crude NDC crystals were washed with a solution of benzene/heptane (60/40) under sonication for 10 min. Pure NDC crystals were collected from the solution by vacuum filtration and dried by storing in vacuum oven at 40 °C for 24 h.

**Solution Polymerization of Copolyesters.** Homogeneous mixture of DMAP (0.04 mole) and CHDM (0.012 mole) in 150 mL of THF was poured into a 250 mL three-necked round-bottom flask, equipped with a magnetic stirrer, a dropping funnel, and a reflux condenser. Transparent solution of equimolar amounts (0.01 mole) of TPC and NDC mixture with the designated ratio was dissolved in 50 mL of THF solvent and added dropwise into the flask through the dropping funnel.



**Scheme 1.** Modification of NDA into NDC by thionyl chloride.

Homogeneous mixtures of all chemicals were mixed by sonication process for 10 min. Diol (CHDM), diacid chloride (mixture of NDC and TPC), and catalyst (DMAP) were used with a molar ratio of 1.2:1.0:4.0. DMAP played a dual role as a catalyst and as an absorbent of by product (HCl). Diacid chloride solution was added dropwise into the flask. A milky white polymer was appeared gradually, indicating that polymer was successfully synthesized. Solution polymerization of polymer took place in 90 min in which the 30 min dropping time of diacid chloride components was included. All the chemical reactions were carried out at room temperature. After completion of reaction, synthesized polymer was precipitated in 600 mL methanol with constant stirring for 30 min. Polymer was purified by washing with methanol and water and it was placed in the vacuum oven for overnight at 50 °C. The schematic diagram of polymer synthesis by solution polymerization reaction is shown in Scheme 2.

Two series of copolyesters based on CHDM and TPC/NDC mixture, PC70TN# and PC100TN# (C = CHDM, numbers after C = mole% of trans content of trans/cis isomer in CHDM, T = TPC, N = NDC, and number after N = mole% of NDC content in diacid unit (TPC/NDC)) copolyesters, were synthesized by the procedures described above. Both series of copolyesters were prepared with different composition of NDC and TPC as summarized in Tables 1 and 2, respectively.

**Characterization.** The actual chemical compositions of the synthesized copolymers were determined by proton nuclear magnetic resonance (<sup>1</sup>H NMR) spectroscopy. <sup>1</sup>H NMR exper-

iments were performed by a Unity Inova 500NB High Resolution 500 MHz NMR Console at 25 °C. Polymer samples were dissolved in a mixture of deuterated chloroform (CDCl<sub>3</sub>) and trifluoroacetic acid (TFA-d) (1:1) and tetramethylsilane (TMS) was used as an internal standard.

Intrinsic viscosity (IV) of all synthesized copolyesters was measured in OCP at 25 °C by an automated ubheload-1C viscometer. Molecular weights (*M<sub>w</sub>*) of synthesized copolyesters were determined using Mark-Houwink equation by following American polymer standards.<sup>38</sup>

Thermal properties like glass transition temperature (*T<sub>g</sub>*),

**Table 1. Composition of Synthesized PC70TN# Copolymers**

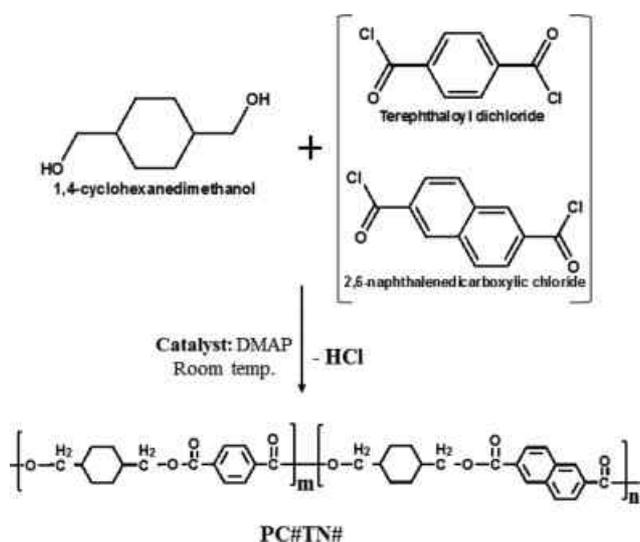
Sample coding	Molar ratio of trans-/cis-CHDM isomer	Molar ratio of feed acids (TPC/NDC)	Molar ratio of acid units (TPC/NDC) <sup>a</sup> in obtained polymer
PC70TN0 (70% trans CHDM)	70/30	100/0	100/0
PC70TN17	70/30	80/20	82.61/17.39
PC70TN26	70/30	70/30	73.17/26.83
PC70TN36	70/30	60/40	63.81/36.19
PC70TN46	70/30	50/50	53.99/46.01
PC70TN55	70/30	40/60	44.80/55.20
PC70TN64	70/30	30/70	35.23/64.77
PC70TN83	70/30	10/90	16.13/83.87

<sup>a</sup>The chemical compositions were determined from the analysis of <sup>1</sup>H NMR spectra.

**Table 2. Composition of Synthesized PC100TN# Copolymers**

Sample coding	Molar ratio of trans-/cis-CHDM isomer	Molar ratio of feed acids (TPC/NDC)	Molar ratio of acid units (TPC/NDC) <sup>a</sup> in obtained polymer
PC100TN0 (100% trans CHDM)	100/0	100/0	100/0
PC100TN17	100/0	80/20	82.48/17.52
PC100TN25	100/0	70/30	74.37/25.63
PC100TN35	100/0	60/40	64.10/35.90
PC100TN46	100/0	50/50	53.47/46.53
PC100TN55	100/0	40/60	44.44/55.56
PC100TN64	100/0	30/70	35.19/64.81
PC100TN83	100/0	10/90	16.64/83.36

<sup>a</sup>The chemical compositions were determined from the analysis of <sup>1</sup>H NMR spectra.



**Scheme 2.** Schematic diagram for the solution polymerization of PC#TN# series.

melting temperature ( $T_m$ ), and the melting enthalpy ( $\Delta H_m$  in J/g) of synthesized copolymers were determined by a differential scanning calorimeter (DSC Q20, TA instruments). About 4-6 mg of the sample was loaded in the DSC sample pan and heated from 40 to 300 °C at a heating rate of 10 °C min<sup>-1</sup> under nitrogen with a purge flow of 50 mL min<sup>-1</sup>. Samples were first heated and cooled down in order to eliminate the thermal history of polymer then values were determined from the second heating cycle.

The thermal degradation behaviors of the synthesized NDC and copolymers were analyzed by a thermo-gravimetric analyzer (TGA Q50, TA instruments). For this, 8-10 mg of the sample was used for the measurement and heated from 40 to 600 °C under nitrogen atmosphere at a heating rate of 10 °C min<sup>-1</sup> with a nitrogen purge flow of 50 mL min<sup>-1</sup>. The temperatures corresponding to 5% and 50% weight loss and the amount of residue (%) at 600 °C were determined from TGA analysis.

## Results and Discussion

**Confirmation of Synthesis of NDC from NDA.** In an effort to improve the reactivity of NDA during solution polymerization, modification of NDA into NDC was attempted and the success of the synthesis was confirmed by DSC and TGA whose results are shown in Figures 1 and 2, respectively.

Melting behavior of resulting NDC was analyzed and compared with starting NDA. Clear melting point at 191.5 °C was observed for NDC while no melting was detected up to 300 °C for NDA. Our results are compatible with the previous reported other researchers data found in the literatures.<sup>37,39</sup> It was found that NDA have superior thermal stability than NDC (Figure 2). From these results, it is concluded that NDA was successfully modified into NDC with a highly efficient reaction (97.7%).

**Determination of Chemical Compositions of Synthesized Copolymers.** Chemical compositions of synthesized PC70TN# and PC100TN# copolyesters were analyzed by <sup>1</sup>H NMR and the results are shown in Figures 3 and 4, respectively. Since obtained copolyesters contain rigid aromatic structures, they were not completely soluble in CHCl<sub>3</sub> so a mixture of TFA-d and CDCl<sub>3</sub>-d (1:1) was used to dissolve the polymer samples. The characteristic chemical shifts ( $\delta$ ) (ppm) peaks at 1.21-2.30, 4.28-4.49, and 4.82 ppm are assigned to hydrogen atoms of 1,4-CHDM diol component of copolymers. The characteristic  $\delta$  peaks at 3.37-3.81 are assigned to hydrogen atoms of -CH<sub>2</sub>OH at chain end of polymer chain. The characteristic  $\delta$

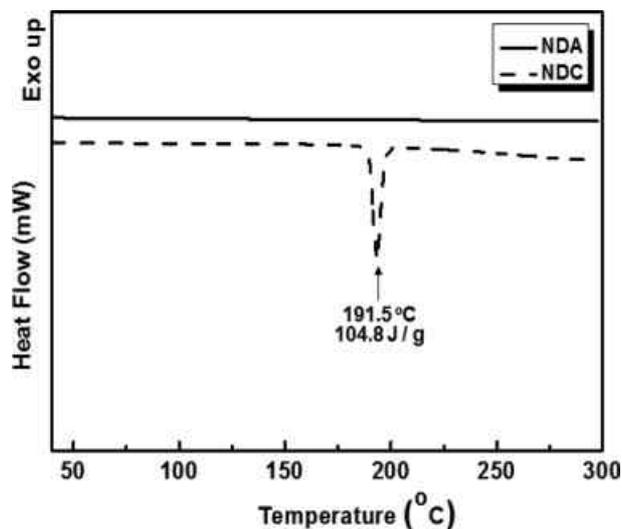


Figure 1. DSC thermograms of NDA and NDC.

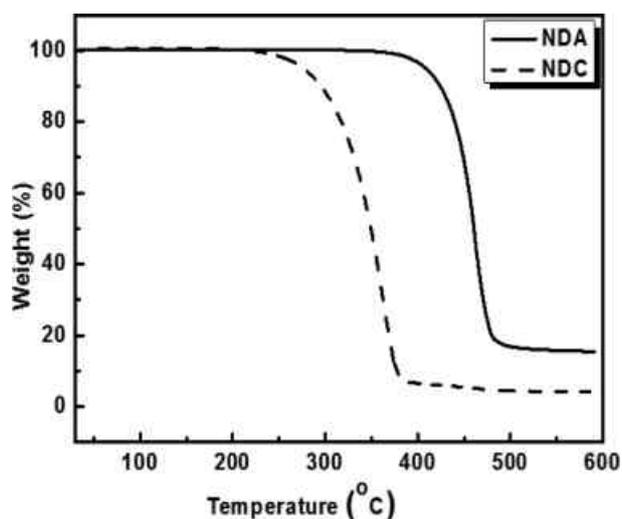
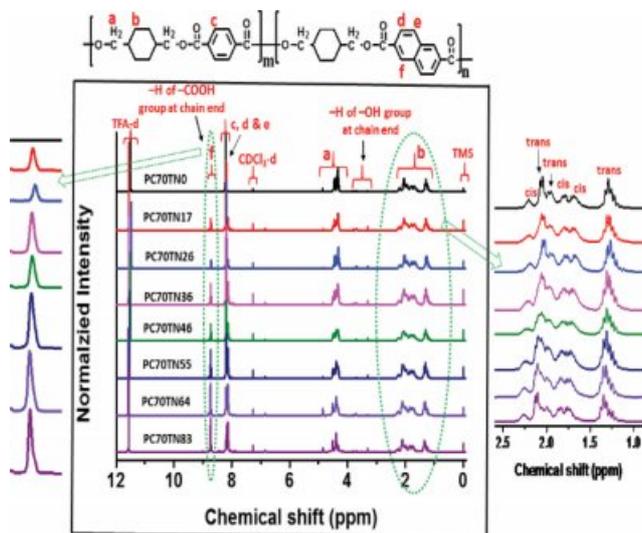
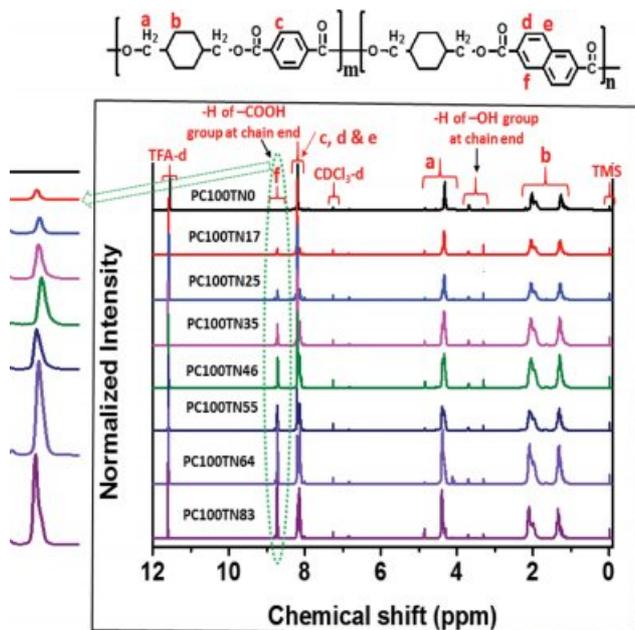


Figure 2. Thermal degradation behaviors of NDA and NDC.

peaks at 8.10-8.18 ppm are assigned to hydrogen atoms of diacid components of copolymers coming from TPC (c) and NDC (d and e), however, the  $\delta$  peak at 8.71 ppm represented by (f) in the figures is assigned to hydrogen atom of NDA at chain end, which enables us to calculate actual chemical compositions of all copolymers. It is evident from the figures that intensity of  $\delta$  peak at 8.71 ppm which represents the naphthalene content in the synthesized copolyesters is increased linearly with increasing the content of NDC in the feeding diacid mixture. In the case of PC70TN0 and PC100TN0, no  $\delta$  peak was observed at 8.71 ppm, which confirms that these copolyesters were not containing NDC. In addition, in the case of PC70TN# series, characteristic  $\delta$  peaks of cis-CHDM observed



**Figure 3.**  $^1\text{H}$  NMR spectra of solution polymerized PC70TN# copolyesters with peak assignments.



**Figure 4.**  $^1\text{H}$  NMR spectra of solution polymerized PC100TN# copolyesters with peak assignments.

at 1.5–1.7, 2.1, and 4.7 ppm were clearly detected while, in the case of PC100TN# series, no peaks were observed. The results of chemical composition analysis for PC70TN# and PC100TN# copolymer series are summarized in Tables 1 and 2, respectively.  $^1\text{H}$  NMR analysis of synthesized copolymers clearly indicates that higher amount of TPA and lower amount of NDA were present in the synthesized PC70TN# and PC100TN# copolymers when compared with feeding amount of mono-

**Table 3.** Molecular Weights and Intrinsic Viscosities of Synthesized PC#TN Copolymers

Trans/cis-CHDM isomer (mole%)	Intrinsic viscosity (g/dL) (IV)		Molecular weight ( $M_n$ )	
	70/30	100/0	70/30	100/0
PCTN0	0.50	0.51	25381	26065
PCTN17	0.48	0.46	24035	22720
PCTN26	0.49	0.50	24704	25381
PCTN36	0.52	0.51	27107	26618
PCTN46	0.48	0.50	24569	25859
PCTN55	0.46	0.43	22332	20000
PCTN64	0.47	0.44	23242	20682
PCTN83	0.49	0.50	25313	25722

mers. This result suggests that the reactivity of NDC is slightly lower than TPC. Our results are compatible with the results of other researchers reported in the literatures.<sup>11,15,23,40</sup>

Molecular Weights ( $M_n$ ) and Intrinsic Viscosity (IV). Intrinsic viscosity (IV) and molecular weights ( $M_n$ ) of obtained copolyesters recorded using Mark-Houwink equation are summarized in Table 3. It important to note that PCTN copolyesters have high enough molecular weights which make them suitable to be used in the form of films, nanofibers or any other desired shape for the practical applications in versatile areas.

**Thermal Properties Examined by DSC.** Solid state polycondensation (SSP) which is normally performed between the  $T_g$  and  $T_m$  has numerous potential advantages to mechanical properties of finally obtained product and it effectively improves the performance properties of polymer. In industry, molecular weight of synthesized polymers is improved by SSP which effectively enhance the mechanical, thermal, and barrier properties of polymer, suitable for wide range of commercial applications.<sup>41,42</sup> In our study, all the results of synthesized PC#TN# copolyesters are analyzed after SSP. Thermal properties of synthesized PC70TN# and PC100TN# copolyesters were analyzed by DSC and the results for  $T_g$ ,  $T_m$ ,  $\Delta H_m$ , and degree of crystallinity are summarized in Tables 4 and 5, respectively.

The degree of crystallinity ( $X_c$ ) of samples was calculated as follows:

$$X_c = \Delta H_m(\text{exp}) \div \Delta H_m(\text{cal}) \times 100\% \quad (1)$$

where  $\Delta H_m(\text{exp})$  is the melting enthalpy determined by DSC experiment and  $\Delta H_m(\text{cal})$  is the melting enthalpy of 100% crystalline PCT which is calculated by adopting a group con-

**Table 4. Summary of Thermal Properties of PC70TN# Copolymers after SSP**

Samples	$T_g$ (°C) <sup>a</sup>	$T_m$ (°C) <sup>a</sup>	$\Delta H_m$ (J g <sup>-1</sup> ) <sup>a</sup>	$X_c$ (%) <sup>b</sup>
PC70TN0	83.8	285.8	48.5	47.6
PC70TN17	84.8	261.7	31.9	31.3
PC70TN26	85.1	246.8	31.4	30.8
PC70TN36	85.4	235.9	39.5	38.7
PC70TN46	88.6	238.2	35.5	34.8
PC70TN55	92.6	255.2	27.2	26.6
PC70TN64	107.5	278.6	28.3	27.8
PC70TN83	114.3	310.2	39.4	38.6

<sup>a</sup>All the results are obtained from DSC measurements from 2<sup>nd</sup> run cycle (heating → quenching → heating).

<sup>b</sup>Calculated by using  $\Delta H_m(\text{exp})/\Delta H_m(\text{cal})$  and  $\Delta H_m(\text{cal}) = 102 \text{ J/g}$ .

**Table 5. Summary of Thermal Properties of PC100TN# Copolymers after SSP**

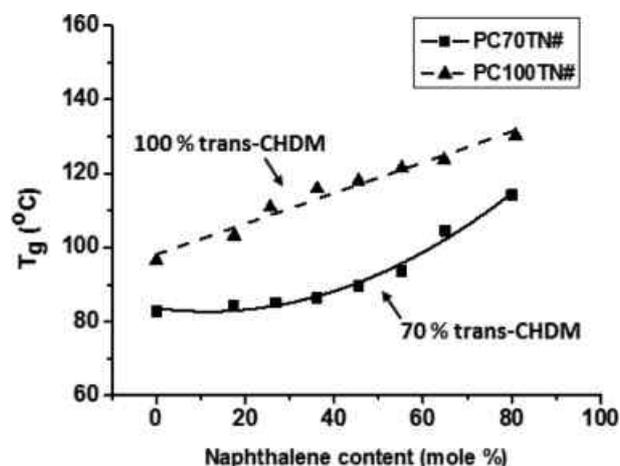
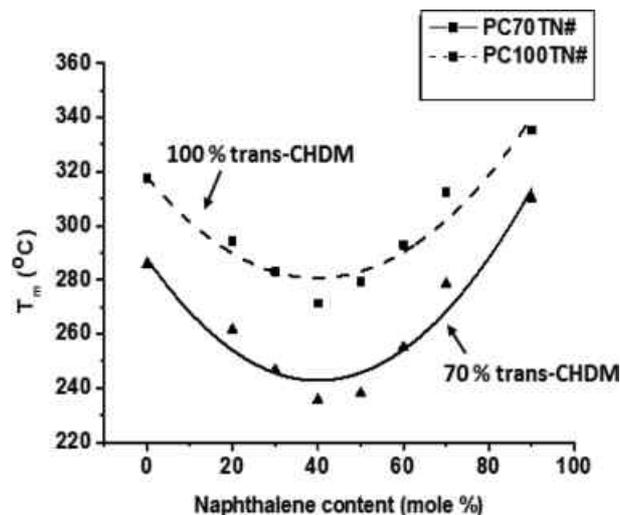
Samples	$T_g$ (°C) <sup>a</sup>	$T_m$ (°C) <sup>a</sup>	$\Delta H_m$ (J g <sup>-1</sup> ) <sup>a</sup>	$X_c$ (%) <sup>b</sup>
PC100TN0	96.4	317.7	65.2	63.9
PC100TN17	103.2	294.4	38.8	38.5
PC100TN25	113.9	285.7	41.9	41.2
PC100TN35	115.9	271.4	41.1	40.2
PC100TN46	118.1	279.3	33.2	32.5
PC100TN55	121.5	292.9	29.6	28.9
PC100TN64	122.6	312.5	42.8	41.9
PC100TN83	130.3	335.3	40.8	39.6

<sup>a</sup>All the results are obtained from DSC measurements from 2<sup>nd</sup> run cycle (heating → quenching → heating).

<sup>b</sup>Calculated by using  $\Delta H_m(\text{exp})/\Delta H_m(\text{cal})$  and  $\Delta H_m(\text{cal}) = 102 \text{ J/g}$ .

tribution theory.<sup>43,44</sup> Inherent thermal properties of synthesized copolyesters were determined after eliminating the thermal history by applying heating-cooling cycle. All the DSC results were determined during the second heating cycle.

To demonstrate the effect of naphthalene content (mole%) on the  $T_g$  and  $T_m$  of PC70TN# and PC100TN# copolymers more clearly, the data shown in Tables 3, 4 are plotted and the results are shown in Figures 5 and 6, respectively. It is evident from the graph that  $T_g$  of both series, PC70TN# and PC100TN#, is increased linearly by increasing the naphthalene units. It is also clear that  $T_g$  of synthesized PC100TN# copolymers prepared with high trans-CHDM are higher than their analogous PC70TN# copolymers prepared with 70% trans-CHDM. This behavior can be resulted from the increased symmetry and rigidity of PC100TN# copolymers that contain 100% trans-CHDM isomer compared to PC70TN# series.

**Figure 5.** Effect of naphthalene content (mole%), defined as  $n/(m+n)$  of Scheme 1, on glass transition temperature ( $T_g$ ) of PC70TN# and PC100TN# copolymers.**Figure 6.** Effect of naphthalene content (mole%), defined as  $n/(m+n)$  of Scheme 1, on melting temperature ( $T_m$ ) of PC70TN# and PC100TN# copolymers.

Melting temperatures of synthesized two series, PC70TN# and PC100TN# copolyesters were also determined by DSC and the results are given in Figure 6 and summarized in Tables 3 and 4, respectively. It is found that  $T_m$  of naphthalene modified PC70TN# and PC100TN# copolymers decreases until the content of naphthalene unit reaches 36 mole% (eutectic point) and then it starts to increase rapidly by increasing naphthalene content furthermore.

At eutectic point, poly(cyclohexane 1,4-dimethylene naphthalate) (PCT) and poly(cyclohexane 1,4-dimethylene terephthalate) (PCN) crystals coexist in the copolymers and after this point the main crystal structure is dominated by the PCN-

type crystals that enhance the thermal and physical properties of copolymers. Similar trends have been reported previously for naphthalene units containing copolyesters which were synthesized by melt polymerization.<sup>15,45</sup> Copolymers synthesized from aromatic diacids and cycloaliphatic trans-CHDM diol have symmetry and rigidity in their molecular structure. It is considered that copolymers based on trans-CHDM have superior thermal properties than their analogous copolymers having benzene based diols components, like 1,4-benzenedimethanol.<sup>46</sup>

It is evident from these data that naphthalene units effectively improve the heat stability of synthesized copolyesters. Not only naphthalene unit but stereochemistry of 1,4-CHDM (trans/cis content) also improve the comprehensive properties of copolyesters.<sup>17-21</sup> Most important finding of this is that PC100TN# copolyesters have superior thermal properties when compared with PC70TN# copolyesters. This behavior can be attributed to trans-CHDM isomers which impart more symmetry and rigidity to their molecular structure.

It can be seen from the Figures 5 and 6 that PC70TN# copolyesters have  $T_g$  ranging between 83.7-114.3 °C and  $T_m$  ranging between 285.7-310.2 °C while PC100TN# copolyesters have relatively superior  $T_g$  and  $T_m$  ranging between 96.4-130.3 °C and 317-335 °C, respectively. Unexpectedly high heat resistance, superior  $T_g$ , and  $T_m$  of PC100TN# copolyesters containing 100% trans-CHDM content can be attributed to the thermally stable and symmetric structure of cycloaliphatic trans-CHDM isomers which facilitate the formation of stable crystal structure.<sup>18,21-25</sup> Similar trends have already been reported for melt polymerized PCT and PCN homopolymers containing high trans-CHDM isomers.<sup>9,23</sup> Therefore, it is concluded that not only the naphthalate content but also the trans-1,4-CHDM isomer impart rigidity and symmetry to copolyester backbone.

**Thermal Stability of Synthesized Copolyesters.** Thermal stability of synthesized copolyesters was analyzed by TGA and the results are given in Figures 7 and 8. Thermograms of only three different compositions of synthesized copolyesters are shown for the clarity of interpretation because the data overlap each other too much when drawn together. However, the detailed results are summarized in Tables 6 and 7. It was found that series of PC70TN# and PC100TN# copolyesters were stable up to 360 °C. Initial degradation temperature at which 5 wt% loss ( $T_{id5\%}$ ) and temperature at which 50 wt% loss ( $T_{id50\%}$ ) of initial weight observed are summarized in Tables 6 and 7 along with the amount of residue at 600 °C. All the copolyesters showed one-stage decomposition process (Figures 7 and 8), indicating that all the synthesized polymers

were random copolyesters. From these figures it is clear that naphthalene units improve the thermal stability of synthesized copolyesters and this effect becomes more prominent when naphthalene content is higher than 46 mole%. Naphthalene units incorporated in copolyesters are thermally more stable than their analogous polyesters containing terephthalate units only. It was found that more residues were observed when naphthalene contents were increased. This finding is comparable to the results reported for the melt polymerized polyesters.<sup>47</sup> Our results show that thermally stable copolyesters can be synthesized by a straightforward one-step solution polymerization process at room temperature in relatively short reaction time.

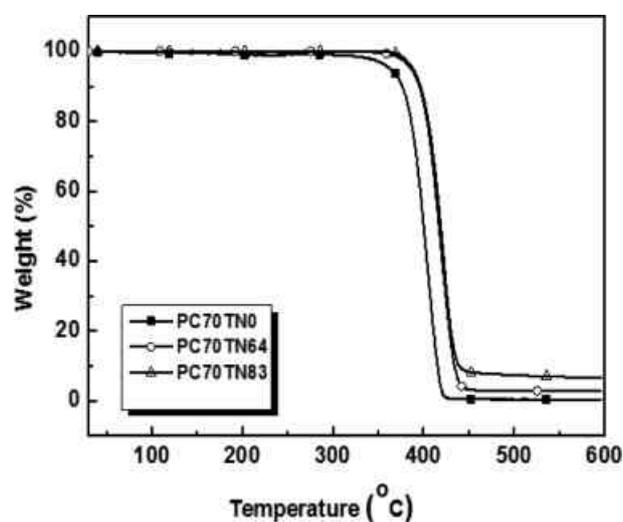


Figure 7. TGA thermograms of synthesized PC70TN# copolyesters.

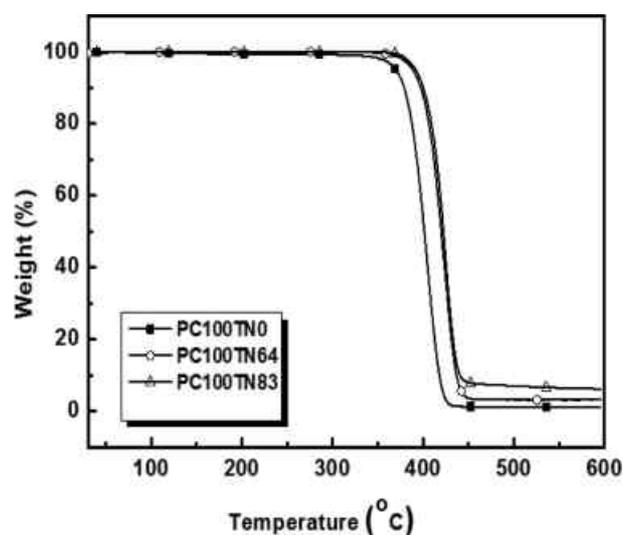


Figure 8. TGA thermograms of synthesized PC100TN# copolyesters.

**Table 6. Thermal Degradation Properties of PC70TN# Copolymers after SSP**

Samples	$T_{id5\%}$ (°C)	$T_{id50\%}$ (°C)	Residues at 600 °C (%)
PC70TN0 (70% trans CHDM)	363.10	400.20	0.40
PC70TN26	362.57	400.01	0.07
PC70TN36	363.01	400.58	1.67
PC70TN46	363.53	400.73	1.75
PC70TN55	364.08	401.06	1.84
PC70TN64	386.52	416.96	2.83
PC70TN83	388.16	418.88	6.52

**Table 7. Thermal Degradation Properties of PC100TN# Copolymers after SSP**

Samples	$T_{id5\%}$ (°C)	$T_{id50\%}$ (°C)	Residues at 600 °C (%)
PC100TN0 (100% trans CHDM)	369.62	400.88	1.18
PC100TN25	368.05	401.25	1.04
PC100TN35	368.46	402.52	0.89
PC100TN46	369.40	403.03	0.45
PC100TN55	373.01	404.26	0.77
PC100TN64	389.33	419.85	3.17
PC100TN83	391.65	421.62	6.21

## Conclusions

Two series of copolyester, PC70TN# and PC100TN# were successfully synthesized by an efficient one-step solution polymerization and their structure-property relationship was studied. CHDM was used as a diol part and various amounts of NDC and TPC were used as diacid part for the synthesis. In order to study the effect of stereochemistry of CHDM unit, CHDM containing 70% trans- and 100% trans-isomers were employed. The determination of chemical compositions of synthesized copolyesters confirmed that NDC is less reactive than TPC. Thermal analysis showed that obtained copolyesters were semi-crystalline in nature and both the content of trans-CHDM isomers and the naphthalene units' amount imparted thermal stability into obtained copolyesters. DSC results revealed that  $T_g$  of both series of copolyesters was increased by increasing the thermally stable naphthalene unit in a linear trend while  $T_m$  was first decreased until eutectic point (36

mole% naphthalene unit) then started to increase rapidly by increasing naphthalene units. It is worthy to note that  $T_g$ ,  $T_m$ , and thermal degradation behavior of PC100TN# copolyesters, containing 100% trans-CHDM isomers were better than their analogues PC70TN# copolyesters that contain 70% trans-CHDM isomers.

To the best of our knowledge, it is the first time that one-step solution polymerization of aromatic PCTN copolyesters containing CHDM, terephthalate, and naphthalate units is conducted at room temperature in the absence of metallic catalyst and stabilizer. Our results also indicate that aromatic homopolyesters (PCT and PCN whose synthesis are very difficult by conventional melt polymerization) can be successfully synthesized by solution polymerization in a relatively short time.

**Acknowledgments:** The authors would like to express appreciations to BASF for the financial support.

## References

1. C. S. Boland, U. Khan, G. Ryan, S. Barwich, R. Charifou, A. Harvey, C. Backes, Z. Li, M. S. Ferreira, M. E. Mobius, R. J. Young, and J. N. Coleman, *Science*, **354**, 1257 (2016).
2. B. K. Behera and H. Arora, *J. Ind. Text.*, **38**, 205 (2009).
3. W. Shen, X. Zhang, Q. Huang, Q. Xu, and W. Song, *Nanoscale*, **6**, 1622 (2014).
4. S. R. Turner, *J. Polym. Sci., Part A: Polym. Chem.*, **42**, 5847 (2004).
5. J. E. McIntyre, in *Modern Polyesters: Chemistry and Technology of Polyesters and Copolyesters*, J. Scheirs and T. E. Long, Eds., Wiley Susse, England, pp 1-28 (2004).
6. C. Bastioli, M. Foa, G. Floridi, F. Farachi, and T. Milizia, US Patent 6,727,342 B1 (2004).
7. W. H. Carothers and J. W. Hill, *J. Am. Chem. Soc.*, **54**, 1557 (1932).
8. J. R. Whinfield and J. T. Dickson, US Patent 2,465,319 (1949).
9. C. J. Kibler, A. Bell, and J. G. Smith, *J. Polym. Sci., Part A: Gen. Pap.*, **2**, 2115 (1964).
10. S. R. Turner and R. W. Seymour, in *Modern Polyesters: Chemistry and Technology of Polyesters and Copolyesters*, J. Scheirs and T. E. Long, Eds., Wiley Susse, England, pp 267-288 (2004).
11. J. M. Koo, S. Y. Hwang, W. J. Yoon, Y. G. Lee, S. H. Kim, and S. S. Im, *Polym. Chem.*, **6**, 6973 (2015).
12. I. N. Duling, US Patent 2,547,730 (1969).
13. B. Hu and R. M. Ottenbrite, in *Modern Polyesters: Chemistry and Technology of Polyesters and Copolyesters*, J. Scheirs and T. E. Long, Eds., Wiley Susse, England, pp 335-359 (2004).
14. Y. U. Shi and S. A. Jabarin, *J. Appl. Polym. Sci.*, **81**, 11 (2001).
15. S. Park, F. Hussain, S. Kang, J. Jeong, and J. Kim, *Polym. Korea*,

- 42**, 662 (2018).
16. H. Kliesh, M. O. Klein, and B. Kuhmann, US Patent 2012/0196111 A1 (2012).
  17. F. Liu, J. Zhang, J. Wang, X. Liu, R. Zhang, G. Hu, H. Na, and J. Zhu, *J. Mater. Chem. A*, **3**, 13637 (2015).
  18. A. Celli, P. Marchese, L. Sisti, D. Dumand, S. Sullalti, and G. Totaro, *Polym. Int.*, **62**, 1210 (2013).
  19. C. J. Kiber, A. Bell, and J. G. Smith, US Patent 2,901,466 (1959).
  20. J. Wang, X. Liu, Z. Jia, L. Sun, Y. Zhang, and J. Zhu, *Polymer*, **137**, 173 (2018).
  21. A. Celli, P. Marchese, S. Sullalti, C. Berti, and G. Barbiroli, *Macromol. Chem. Phys.*, **212**, 1524 (2011).
  22. C. Berti, A. Celli, P. Marchese, E. Marianucci, G. Barbiroli, and F. D. Credico, *Macromol. Chem. Phys.*, **209**, 1333 (2008).
  23. Y. G. Jeong, W. H. Jo, and S. C. Lee, *Macromolecules*, **36**, 5201 (2003).
  24. B. Vanhaecht, M. N. Teerenstra, D. R. Suwier, R. Willem, M. Biesemans, and C. E. Koning, *J. Polym. Sci., Part A: Polym. Chem.*, **39**, 833 (2001).
  25. D. J. Lyman, *J. Polym. Sci.*, **55**, 507 (1961).
  26. Y. G. Jeong, W. H. Jo, and S. C. Lee, *J. Polym. Sci., Part B: Polym. Phys.*, **42**, 177 (2004).
  27. L. Hu, L. Wu, F. Song, and B. G. Li, *Macromol. React. Eng.*, **4**, 621 (2010).
  28. A. Lodha, R. S. Ghadage, and S. Ponrathnam, *Polymer*, **38**, 6167 (1997).
  29. T. H. Shah, J. I. Bhatta, G. A. Gamlen, and D. Dollimore, *Polymer*, **25**, 1333 (1984).
  30. G. P. Karayannidis, C. P. Roupakias, D. N. Bikiaris, and D. S. Achilias, *Polymer*, **44**, 931 (2003).
  31. W. Shotyky and M. Krachler, *Environ. Sci. Technol.*, **41**, 1560 (2007).
  32. K. Pang, R. Kotek, and A. Tonelli, *Prog. Polym. Sci.*, **31**, 1009 (2006).
  33. R. Muller, *Trans. R. Soc. Trop. Med. Hyg.*, **97**, 124 (2003).
  34. S. L. Kwolek and P. W. Morgan, *J. Polym. Sci., Part A: Gen. Pap.*, **2**, 2693 (1964).
  35. E. D. Giol, N. V. den Brande, B. V. Mele, S. V. Vlierberghe, and P. Dubruel, *Polym. Int.*, **67**, 292 (2018).
  36. B. J. Sublett, US Patent 5,616,404 (1997).
  37. L. Starr, *J. Polym. Sci., Part A-1*, **4**, 3041 (1966).
  38. American Polymer Standards Co., *Polymer Standards Catalog* (1991).
  39. A. Paul, D. A. Young, G. E. Kulhman, W. Partenheimer, and W. P. Schammel, US Patent 4,950,786 (1990).
  40. C. A. Boye, *J. Polym. Sci.*, **55**, 275 (1961).
  41. I. S. Rogulska and G. Rokicki, *Polimery*, **58**, 85 (2013).
  42. N. Kasmi, G. Z. Papageorgiou, D. S. Achilias, and D. N. Bikiaris, *Polymers*, **10**, 1 (2018).
  43. M. Ayyoob, D. H. Lee, J. H. Kim, S. W. Nam, and Y. J. Kim, *Fibers Polym.*, **18**, 407 (2017).
  44. Y. C. Feng, H. Zhao, T. H. Hao, G. H. Hu, T. Jiang, and Q. C. Zhang, *Materials*, **10**, 694 (2017).
  45. Y. G. Jeong, W. H. Jo, and S. C. Lee, *Macromolecules*, **36**, 4051 (2003).
  46. E. V. MaTic and C. J. Kibler, in *Man-made Fibres: Science and Technology*, H. F. Mark, S. M. Atlas, and E. Cernia, Eds., Interscience, New York, pp 83-134 (1967).
  47. I. C. Mcneill and M. Bounekhel, *Polym. Degrad. Stab.*, **34**, 187 (1991).